

## Color Evolution and Thermal Behavior of Silica (SiO<sub>2</sub>)- and Zircon (ZrSiO<sub>4</sub>)-coated Nano $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> Pigments

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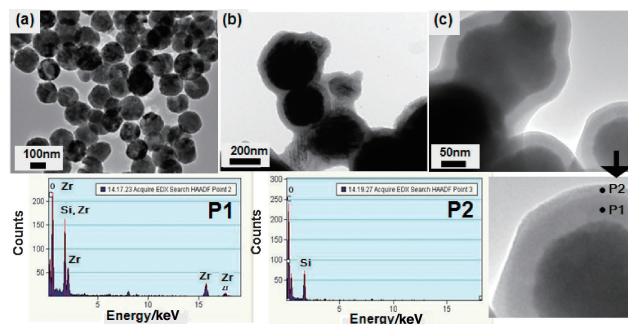
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This manuscript reports coloration and thermal behavior of the silica and zircon effect on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> pigment. The prepared three types of pigments were heat-treated at various temperatures (800–1000 °C), and their color evolutions were characterized by colorimeter and UV–visible spectroscopy.

Red color  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> has attracted much interest owing to recent demand for nontoxic, Cd-free pigments.<sup>1</sup> In particular, the coloration of nanosized  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is controllable depending on the synthetic conditions.<sup>2–4</sup> In general, ceramic pigments must possess thermal stability at high temperature. However, nanosized red pigment easily undergoes aggregation at high temperature, and its color turns to brown. To obtain a stable and bright color, it is necessary to encapsulate  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> in a more stable phase such as silica (SiO<sub>2</sub>) or zircon (ZrSiO<sub>4</sub>). Several groups have studied silica inclusion red pigment in a silica matrix using FeSO<sub>4</sub>·7H<sub>2</sub>O as an iron oxide precursor.<sup>5,6</sup> Recently, well-designed core–shell red pigments were prepared using nanosized  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and silica (SiO<sub>2</sub>) from tetraethyl orthosilicate (TEOS) as starting materials.<sup>7</sup> Silica is particularly attractive for use as a coating and various applications due to its high thermal stability and transparency.<sup>8,9</sup> Another promising encapsulator, ZrSiO<sub>4</sub>, is commonly used in the ceramic industry for glazes due to its high chemical and thermal stability at firing temperature.<sup>10</sup> Zircon (ZrSiO<sub>4</sub>) was sintered from ZrO<sub>2</sub> and SiO<sub>2</sub> layers which are coated separately in a corresponding solution.<sup>11,12</sup> In this work, we propose two well-designed core–shell nanopigments using silica (SiO<sub>2</sub>) and zircon (ZrSiO<sub>4</sub>) prepared by a hydrolysis method and compare the resulting color evolution to that of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles. Sphere-shaped  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles were prepared via a hydrothermal method as described elsewhere.<sup>2,13</sup> Silica-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> was synthesized following a modified version of a method from the literature.<sup>7</sup> The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (40 mg) nanoparticles were dispersed in 100 mL of distilled water by sonication for 5 min followed by the addition of 1 wt % cetyltrimethylammonium bromide (CTAB) (1 mL). Then, 10 mL of (0.02 M) aqueous zirconium sulfate solution was added dropwise to the above solution at room temperature for 2 h. After zirconia (ZrO<sub>2</sub>) shell of about 16–25 nm thick, silica layers (SiO<sub>2</sub>) were fabricated using the Stöber method in an ethanol–water solution of TEOS.  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub>, and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrO<sub>2</sub>@SiO<sub>2</sub> were sintered at 800, 900, and 1000 °C for 1 h at a heating rate of 10 °C min<sup>-1</sup>. The samples were characterized by X-ray diffraction (XRD), transmission electron microscopy (TEM), and energy dispersion spectroscopy (EDS) equipment. Color measurements were performed by UV–vis spectroscopy, using the CIE Lab method.

A typical TEM image (Figure 1a) shows the sphere-shaped  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> particles with an average diameter of 150 nm.

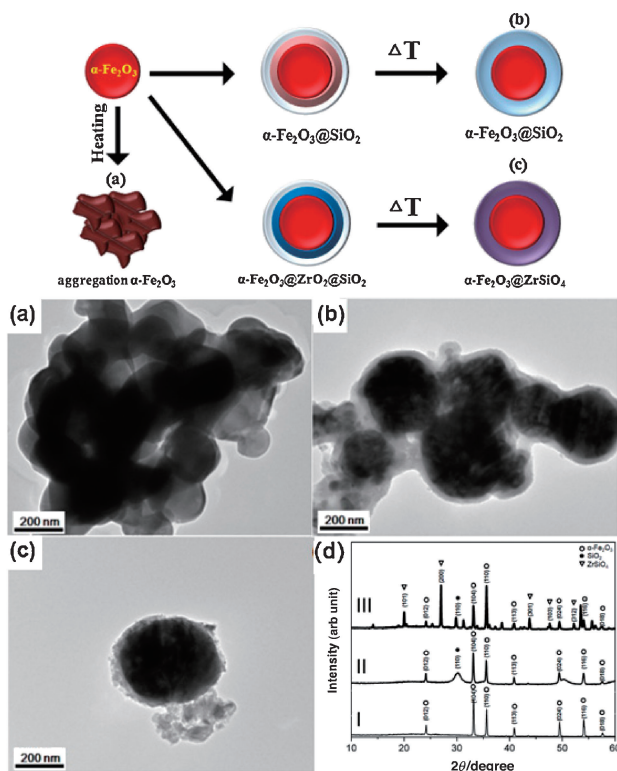


**Figure 1.** TEM images of (a)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles (bar: 100 nm), (b)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub> nanoparticles, (c)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrO<sub>2</sub>@SiO<sub>2</sub> nanoparticles and EDS spectra of spots (P1)–(P2), respectively, in insert image.

Figure 1b shows TEM image of silica-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles that have a silica layer with a thickness of around 45 nm. CTAB was chosen as a surface modifier, and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@CTAB@SiO<sub>2</sub> was successfully prepared. Synthesis of ZrSiO<sub>4</sub>-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is done through sintering from ZrO<sub>2</sub> and SiO<sub>2</sub>-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>.<sup>11,12</sup> The ZrO<sub>2</sub> layer and SiO<sub>2</sub> layer are obtained separately in corresponding solution by hydrolysis. A TEM photograph (Figure 1c) of the as-prepared particles clearly indicates that the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> particles were coated with double shells. The ZrO<sub>2</sub> and SiO<sub>2</sub> shells were about 25 and 9 nm in thickness, respectively. The EDS results in P1 and P2 confirm the formation of ZrO<sub>2</sub> and SiO<sub>2</sub> layers, as indicated by the TEM photograph in Figure 1c.

When the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles were heated to 1000 °C, the result was extensive agglomeration of the nanoparticles to form microsized units (Figures 2a and 2d-I). At the same time, silica-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles retained their shapes after being heated to 1000 °C, as illustrated in Figure 2b. Figure 2d shows the XRD pattern of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub>, and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrSiO<sub>4</sub> after being heated at 900 °C. Unfortunately,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrO<sub>2</sub>@SiO<sub>2</sub> core–shell structure was not found in the XRD pattern at 1000 °C, due to aggregation and melting in the glassy matrix after heating (Figure 2c). In case of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub> at 900–1000 °C, amorphous SiO<sub>2</sub> was crystallized which was indexed and comparable with room temperature data (Figure 2d-II).<sup>7</sup> When samples were treated at 900 °C, the formation of ZrSiO<sub>4</sub> phase was observed in Figure 2d-III. LiF easily reacts with SiO<sub>2</sub> to form SiF<sub>4</sub> during heat treatments which can induce ZrSiO<sub>4</sub> formation at low temperature (<850 °C).<sup>10,14</sup>

The color of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and core–shell structure was assessed on the grounds of  $L^*$ ,  $a^*$ , and  $b^*$  parameters, calculated from the diffuse reflectance spectra. In this method the parameter  $L^*$  represents the brightness of the sample; a

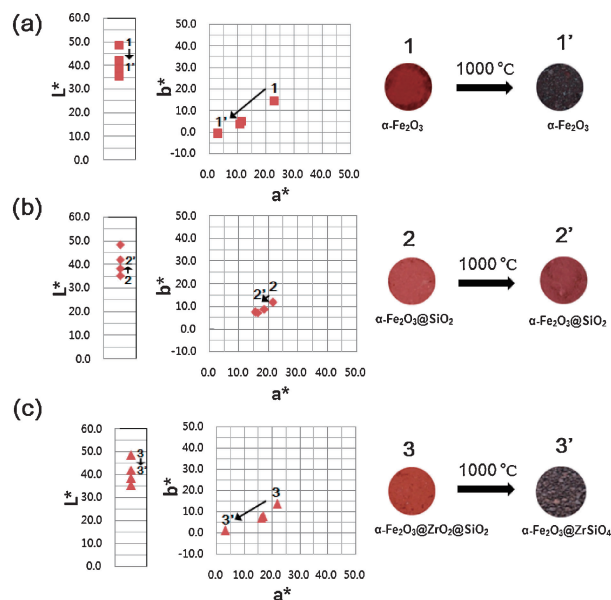


**Figure 2.** TEM images of after heating at 1000 °C (a)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> nanoparticles, (b)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub> nanoparticles, (c)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrO<sub>2</sub>@SiO<sub>2</sub> nanoparticles, and (d) XRD data of three samples at 900 °C (I:  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, II:  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub>, III:  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrO<sub>2</sub>@SiO<sub>2</sub>).

positive  $a^*$  value represents red, a negative value shows green. It is said that red pigments show a high red color tone with increasing  $a^*$ . Silica did not affect red color tonality due to its high transparency. As temperature increases, the  $a^*$  and  $b^*$  factors of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> evidently decreased due to the aggregation and morphological change of nanosized  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> (Figure 3a). At 1000 °C,  $a^*$  and  $b^*$  values of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrO<sub>2</sub>@SiO<sub>2</sub> decreased to near 0 and turned to brownish gray because lithium iron silicate was melted in the glassy matrix (Figure 3c). On the other hand, the constant  $a^*$  and  $b^*$  values in the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub> samples indicate that the level of red pigments was relatively stable. The  $a^*$  value of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub> was about 18, which is comparable to the value found in a spindle-type core structure (Figure 3b).<sup>7</sup>

The color of the prepared powders was characterized by reflectance spectra in the wavelength range from 350 to 800 nm by UV-vis spectroscopy (Figure S1<sup>16</sup>). Figure S1<sup>16</sup> presents the reflectance curves of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub>, and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrO<sub>2</sub>@SiO<sub>2</sub>, respectively. The reflectance value is strongly influenced by the mean diameter.<sup>15</sup> Silica-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> shows higher reflectance in a longer wavelength region (700–800 nm) than the others pigments do. After heat treatment at 900 °C, reflectance of the silica-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> had increased due to high scattering of silica surface which is related to the increased  $L^*$  values in the CIE Lab at 900 °C (Figure 3b).

In summary, we report the well-designed synthesis of core-shell structure;  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub>, and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrO<sub>2</sub>@SiO<sub>2</sub> have been achieved by development of the



**Figure 3.** The CIE  $L^*$ ,  $a^*$ , and  $b^*$  values of the pigments in comparison with the values of (a)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, (b)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@SiO<sub>2</sub>, and (c)  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>@ZrSiO<sub>4</sub> at various temperatures at RT, 800, 900, and 1000 °C. Color evolution data (RT, 1000 °C) were showed (Data at 800 and 900 °C are not indexed for clarity).

hydrolysis method. Especially, zircon-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> particles were coated with a double shell of zirconia and silica. At high temperature,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and zircon-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> particles show large coloration change due to morphological change of nano-sized  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and melting in the glassy matrix. Silica-coated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is highly stable optical properties at 1000 °C.

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